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135-8-1/28

Smelting of basic pig iron with oxygen enriched blast. (Cont.)

of the smelting process during the individual periods operating factors and heat balances for the same periods are given in Tables 4, 5 and 6 respectively. The distribution of CO₂ content in the top gas along the throat diameter during the individual operating periods is shown in Fig.7. Variations in the composition and temperature of gas at various furnace levels during the individual operating periods are shown in Figs.8 and 9. Methods used for the determination of the above data are not given. The comparison of cost of production per ton of pig with normal (A) and oxygen enriched (B) blast is given in Table 7. It is concluded that: 1) operation of the furnace with oxygen enriched blast was stable without increasing moisture content of blast. The temperature of the blast was increased by 35-45 C in comparison with the operation on normal blast; 2) oxygen enrichment permitted intensifying furnace driving within the limits of retaining the amount of gas produced per unit of time on the same level as in normal operation; 3) the distribution of the gas stream across the furnace during operation with enriched blast remained normal which was the main factor contributing to

Card 3/5

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1.33-8-1/28

Smelting of basic pig iron with oxygen enriched blast. (Cont.)

the retention of a comparatively low coke rate achieved on normal blast; 4) blast enrichment causes a decrease in the size of zone of moderate temperatures (800-1100 C) in the furnace stack and the corresponding increase in zones of low and high temperatures (above 1100 C). Despite this, the concentration of CO₂ in gas increases along the furnace height at a higher rate than with ordinary blast, due to an increase in the reducing ability of the gas; 5) with a 23.3% oxygen enrichment the output of the furnace increased by 6.7% with unchanged coke rate (14 days operating period); 6) the cost of production of pig with oxygen enriched blast was 2 roubles 40 kop.per ton higher than with ordinary blast. This increase was caused by the high cost of technical oxygen on the HTMK (15 kop/m³); 7) the results obtained fully justify an enlargement of the tonnage oxygen plant on the HTMK in order to supply blast furnaces with oxygen for blast enrichment. Oxygen plants should be built on works operating with a prepared burden. The construction of oxygen plants of 10 000 m³ capacity is recommended.

Card 4/5

Smelting of basic pig iron with oxygen enriched blast. (Cont.)

There are 7 tables, 9 figures and 1 American reference.

ASSOCIATION: TSNIIChM and NizhmyTagil, Metallurgical Combine. (TSNIIChM i Nizhne-Tagil'skiy Metallurgicheskiy Kombinat).

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ZAKHAROV, A.F.

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AUTHOR: Zakharov, A.F., Khil'kevich, F.A., Bazilevich, S.V. and Lazarev, B.L., Engineers.

Smelting of Ferro-manganese in a Large Blast Furnace TITLE: (Vyplavka ferromargantsa v bol'shoy domennoy pechi)

PERIODICAL: Stal', 1957, No.7, pp. 580 - 584 (USSR)

ABSTRACT: In 1956, the smelting of ferro-manganese was carried out in a large furnace (No.2 furnace Nizhne Tagil'skiy Works) (1 100 m) with high top pressure (0.5 atm.) and cxygenenriched blast (up to 24.5%). The preparation of the furnace for the transfer from foundry iron to ferro-mangamese production, characteristic of raw materials, operational practice and the results obtained are described. The profile of the furnace and the distribution of CO2 in the top gas along the

throat diameter are shown in Figs. 1 and 2, respectively. Material and heat balances are given in Tables 1 and 2, res-The comparison of main indices of heat balances of pectively. smelting ferro-manganese in three different works is given in Table 3. In addition, the distribution of temperatures and changes in the gas composition along the height of the furnace stack (Fig. 3) and the composition of gas in the combustion Card 1/2zone (Fig. 4) were studied. It is concluded that on smelting

ferro-manganese in a large Blast Furnace. 133-7-2/28

ferro-manganese in a large furnace using acid slags (810₂ 31.5%, Al₂O₂ - 13.7%, CaO - 31.4%, MgO - 4.1%, MnO - 18.2) and
high temperature blast (998°C) on even furnace operation at a
comparatively high driving rate (blast volume 1 410 m'/min;
ore) with a low coke rate (1 424 kg/ton) can be obtained withoxygen enrichment of blast did not require an increase in its
ferro-manganese under the above conditions; it is necessary to
observe the horizontal distribution of materials across the
that blast temperatures above 1 COO C cam be used. Smelting
in servicing slag notches, tuyere equipment and tapping hole.

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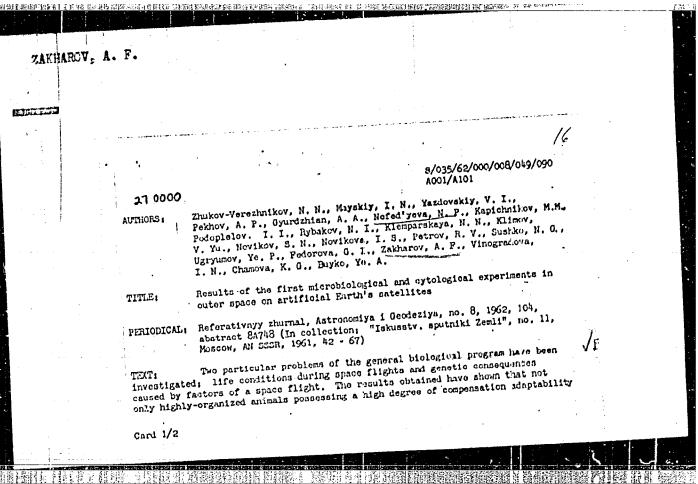
133-7-2/28

131-5%, MgO - 4.1%, MgO - 4

ASSOCIATION: N:zhne-Tagil'skiy Metallurgical Combine (Nizhne-Tagil'skiy Metallurgicheskiy Kombinat)

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CHISTYAKOV, A.H.: YEVDOKIKOV, Yu.P., ZAKHAROV, A.F. [deceased]

Properties of pitches and distillates in a three-stage exidation process. Trudy LTI no.51:159-163 '59. (MIRA 13:8) (Pitch) (Oxidation)

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

ZAKHAROV, A.F.

PHASE I BOOK EXPLOITATION

SOV/4601

- Koordinatnoye soveshchaniye po primeneniyu kisloroda na metallurgicheskikh zavodakh Urala. Sverčlovsk, 1956
- Primeneniye kisloroda na metallurgicheskikh predpriyatiyakh Urala; materialy koordinatsionnogo soveshchaniya (Use of Oxygen in Metallurgical Plants of the Urals; Materials of the Coordination Conference) Sverdlovsk, 1960. 152 p. Errata slip inserted. 1,000 copies printed.
- Sponsoring Agencies: Akademiya nauk SSSR. Ural'skiy filial. Institut metallurgii; Ural'skiye pravleniya nauchno-tekhnicheskikh obshchestv chernoy i tsvetnoy metallurgii.
- Resp. Ed.: P.S. Kusakin, Candidate of Technical Sciences; Tech. Ed.: N.F. Seredkina.
- PURPOSE: This collection of papers is intended for scientific research and technical personnel in the field of metallurgy.
- COVERAGE: The use of oxygen in ferrous and nonferrous metallurgy of the Urals is discussed. Results of experimental use of oxygen in some metallurgical plants are presented. During the Conference, held December 20 and 21, 1956, the following persons (in addition to the authors) took part in Card 1/5

Use of Oxygen (Cont.)

S01/4601

CIA-RDP86-00513R001963520005-0"

the discussion: V.Ya. Miller, V.V. Mikhaylov, P.Ya. Sorokin, A.A. Perestoronin (all affiliated with the Institute of Metallurgy of the Ural Branch AS USSR), S.M. Kazachenko (Nizhne-Saldinskiy metallurgicheskiy zavod - Nizhnyaya-Salda Metallurgical Plant), M.F. Kochin (Deceased) (Ural'skiy institut chernykh metallov - Ural Institute of Ferrous Metals), M.Ye. Kislitsin (Chelyabinskiy metallurgicheskiy zavod - Chelyabinsk Metallurgical Plant), C.V. Demin (Krasnoural'skiy medeplavil'nyy zavod - Krasnoural'sk Copper Smelting Plant), V.A. Aglitskiy (Institut Unipromed' - "Unipromed'" Institute). Some of the papers are followed by references, both Soviet and non-Soviet.

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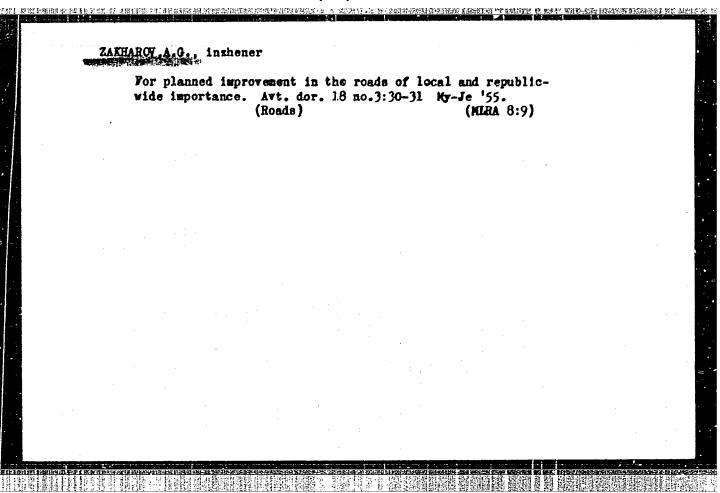
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ZAKHAROV. A. G.

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ZAKHAROV, A. G.

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The booklet is intended for students of polygraphic technical schools.

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计图像特性记行转差 电音点集 工程的资源基本资本的主义的特殊工术的第三人称单数加强调整的高级数据编纂 医囊肿病 的复数数数据 医动物

SOKOLOV, A.A., kand.tekhn.nauk; ZAKHAROV, A.G., inzh.; VASIL'YEV, V.I., inzh.

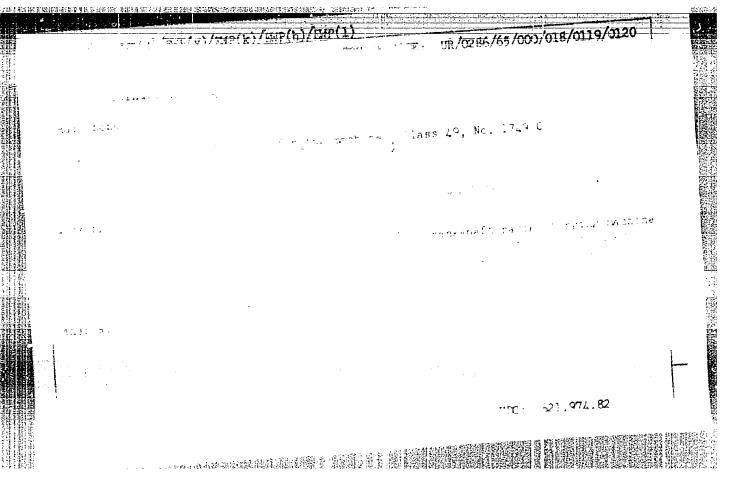
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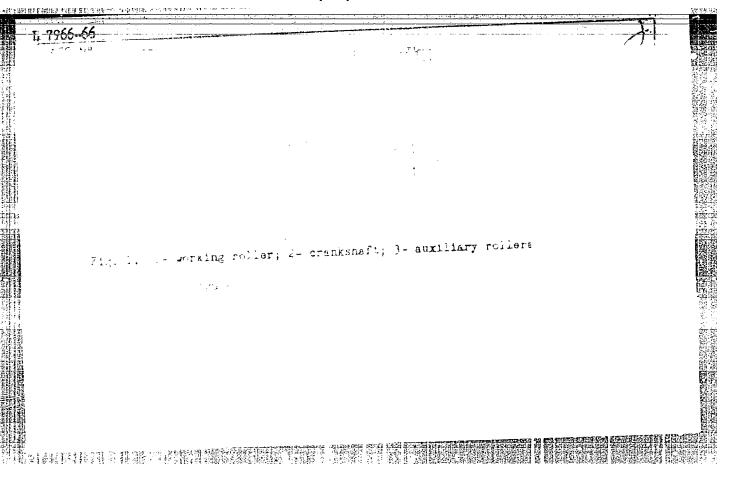
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APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

CIA-RDP86-00513R001963520005-0 'APPROVED FOR RELEASE: 03/15/2001

Zakharov, A.I.

6-1-6/16

AUTHOR:

Zakharov, A. I.

TITLE:

Two-Component-Lens-Compensators With Double Curvature (Dvukhkomponentnyy linzovyy kompensator dvoyakoy krivizny)

Geodeziya i Kartografiya, 1958, Nr 1, pp. 47 - 50 (USSR)

ABSTRACT:

PERIODICAL:

The range-finder attachment AHB-2 (manufactured by the plant "Aerogeopribor") was largely used during recent years. The description of the compensator fixed in this attachment and the dividing device is contained in the elaborate investigation by P. I. Durneva (Geouzdat Publishing house, 1953). It is shown that for increasing the accuracy of distance-measurements by means of this range-finder, the accuracy of the measurement of the parallax angle increases and a surveyor's rod of greater length should be used. Further it is shown that an increase of the accuracy of measurement of line lengths can only be obtained by a modification of the construction of the attached device on the range-finder, especially by changing the main part of the same, viz. the compensator. In 1956, the manufacturers elaborated a new design of a compensator and manufactured experimental types of a range-finder attach-

Card 1/3

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

6-1-6/16

Two-Component-Lens-Compensators With Double Curvature

ment Π HT to the theodolite TT-50. This compensator is a two--component-lens-compensator with double curvature. From the scheme of the compensator given here it results that due to such a scheme it was made possible to combine the front semi--lenses in a common mounting. Moreover, these front semi--lenses can be displaced with respect to the semi lenses in the rear, by which both pictures of the object are displaced in opposed directions. In this case a parallax is missing between the pictures, as well as a difference in the enlargement of the two pictures, since the distance between the principal planes of the components equals zero. It is shown that the range-finder AHT makes it possible to use a two meter surveyor's rod with measuring distances over 200 m, whereas with working with one meter rods only the half of such distances can be measured. Due to the simultaneous displacement of both picutres of the rod marks, the same coincide each time with the measurement of the double parallax angle exactly in the center of the field of view. When measuring the single parallax angle, they are found symmetrically to the center, however, (with the second coincidence - exactly in the center). Consequently, the construction of the new compensator satis-

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APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

Two-Component-Lens-Compensators With Double Curvature

6-1-6/16

fies the basic requirements of increasing the accuracy of measurement. There is 1 figure and 1 non-Slavic reference.

AVAILABLE:

Library of Congress

Card 3/3

ZAKHAROV, A.I.

AUTHOR8:

807/6-58-9-3/26

TITLE:

The Novel Thirty-Second Transit TT5 (Novyy tridtsati-

sekundnyy teodolit TT5)

PERIODICAL:

Geodeziya i kartografiya, 1958, Nr 9, pp 18 - 27 (USSR)

ABSTRACT:

This is a description of the new type of transit, which is to replace old TT50, which is no longer produced. This new transit tachymeter TT5 is designed to measure horizontal and vertical angles with a mean square deviation better than + 15" in one run and to determine distances by means of the cross-hair range-meter. The

new transit is lighter by 2 kg as compared to the old one. With legs it weighs 3,2 kg. A description of attachments furnished on request is presented: The range-meter set DNT - 2 for measuring distances from 50 to 700 m with a mean square deviation of 1:500, the range-meter set DD Z for measuring distances from 20 to 200 m with a mean square deviation of 4:2000, the optical centering device OTs-2x, the compass with an azimuthal circle

Card 1/2

BKT and a set of electrical attachments KEO for work

CIA-RDP86-00513R001963520005-0" APPROVED FOR RELEASE: 03/15/2001

The Novel Thirty-Second Transit TT5

SOV/6-58-9-3/26

at dusk and at night. This paper also includes a description of a variant of the TT5, the transit-tachymeter TTP. This instrument is used in the measurement of horizontal and vertical angles, in the determination of distances with the cross-hair range-meter, and, in combination with the attachments ' DNT-2 and TOZ in the measurement of azimuthal angles. It can be used in leveling work and in the accurate measurement of sightings with a great angle of inclination. Apart from this instrument the level transit TT5 was developed for town surveying and engineering surveys on the initiative of the Mosgorgeotrest, which is tased upon the same transit TN. A short description of this instrument is included in this paper. Finally, results from the testing of the three new instruments are presented. There are 10 figures and 1 reference, which is Soviet.

Card 2/2

24.3300 (1051,1057,1163)

32686 8/035/61/XXX/012/040/043 A001/A101

AUTHORS:

Durneva, P.I., Zakharov, A.I., Kolkov, D.D.

TITLE:

New geodetic instruments: TOM(TOM) theodolite and 耳耳5 (DD5) range

finder

Referativnyy zhurnal. Astronomiya i Geodeziya, no. 12, 1961, 40, FERIODICAL: abstract 12G259 ("Geod. i kartografiya", 1961, no.8, 37 - 47)

The authors describe the small TOM optical theodolite and the DD5 TEXT: differential range finder (attachment) manufactured in serial production in the USSR since 1960. The results of their investigation carried out by TsNIIGAi.K are presented. The main technical characteristics of the theodolite are as follows: magnification of the visual telescope is 18x, visual field is 20, the optical diameter of the objective is 27 mm, diameter of exit pupil is 1.5 mm, equivalent focal length of the objective is 142.5 mm, minimum sighting distance

is 2 m, diameters of the horizontal and vertical circles are 70 mm each, the least scale interval on the circles is 10', magnification of the reading micro. scope is 27x, precision of reading on the circles (ocular estimation) is 1', the scale interval on the level of the horizontal circle alidade is 45" per 2 mm,

Card 1/3

32686 8/035/61/000/012/040/043 A001/A101

New geodetic instruments ...

the scale interval on the level at the telescope is 30" per 2 mm. The weight of the theodolite in a metallic box is 3.2 kg. The visual telescope of the theodolite is anallactic with inner focusing. The telescope objective has three lenses, it is non-glued. The reticule has range finding dash lines; coefficient of the range finder is 100. A cylindrical level is fastened on the visual telescope, which enables one to perform leveling with the horizontal ray. The theodolite is equipped with a round dismountable compass. All main parts of the instrument are manufactured of light and durable alloys. A lens compensator is used in the DD5 range finder, the constant parallactic angle is equal to 17'12"3 (coefficient of the range finder is 200). The operational principle of the range finder is the same as in DD2 and DD3 range finders (of. RZhAstr, 1959, no. 7, 5844, no. 11, 8650). The DD5 range finder is intended for measuring distances 40-200 m with a vertical rod. The rod is two-sided, 1.5 m long, divisions are made on a stretched invar band. In measuring distances from 40 to 160 m, the roi side with 2-cm divisions is used, whereas in measuring distances from 100 to 200 m the side with 5-cm divisions is used. It was found as a result of investigating two TCM theodolites: mean-square error in measuring a direction by one observation (distances to sight targets 1 - 3 km) was ± 0.22 - 0.29; mean-square error of a horizontal angle measured by the method of circular observations was $\pm 0.3 - 0.4$, divergences

Card 2/3

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New geodetic instruments ...

in angle values in different observations did not exceed 1', misclosures in triangles were \sim 1' (maximum 1'.8), mean-square error in measuring a vertical angle by one observation was \pm 0.4 - 0'.7, mean relative error in determining distance with a filament range finder was 1:300 - 1:400, error in leveling by horizontal ray (at the length of sight ray 100 m) was \pm 22 mm/km. Time consumption for observations of 5 directions, once for each, amounts to 4 min, and for measuring a horizontal angle by one observation 1.3 min. Precision of measuring distances from 48 to 200 m with the DD5 range finder (at inclination angles 0-33°) is characterized by mean-square relative error of the order of 1:1,200 - 1:1,600. No more than 1 min is spent for measuring a distance and a vertical angle.

V. Sinyagina

[Abstracter's note: Complete translation]

Card 3/3

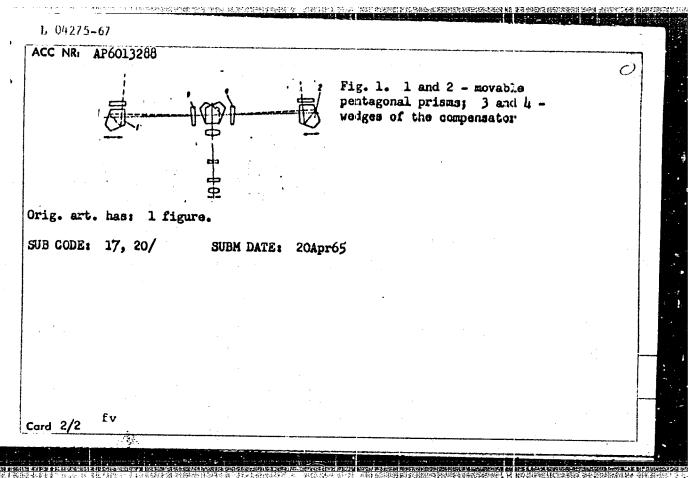
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ZAKHAROV, Anatoliy Ivanovich; ZUYKOV, Ivan Ivanovich; YELISEYEV, S.V., red.

[Medium-precision theodolites and optical telemeters] Teodolity srednei tochnosti i opticheskie dal'nomery. Moskva, Nedra, 1965. 171 p. (MIRA 19:1)

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

L 04275-67 EWT(d) ACC NR: AP6013288 (A) SOUNCE CODE: UR/0413/66/000/008/0084/	37
AUTHOR: Zakharov, A. I.	B
ORG: none TITLE: Self-correcting double-image range finder. Class 42, No. 180814 SOURCE: Izobreteniya, promyshlennyye obraztsy, tovarnyye znaki, no. 8, 1966, TOPIC TAGS: optic range finder, optic instrument, optic lens ABSTRACT: This Author Certificate presents a self-correcting double-image ra ABSTRACT: This Author Certificate presents a self-correcting double-image ra finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of the objective of the view finder with a variable instrument base. The latter is formed of the objective of the view finder with a variable instrument base. The latter is formed of the objective of the view finder with a variable instrument base. The latter is formed of the objective of the view finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables finder with a variable instrument base. The latter is formed of two movables for the variable instrument base. The latter is formed	wer 1). listances
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ZAKHAROV, A.I.; KAROMISEVA, A.M.; IEVITIN, R.Z.; PONYATOVSKIY, Ye.C.

Magnetic and magnetoelastic properties of the metric of alloy from shodium. When eksp.i teor.fiz. 46 no.6:2003-2010 Jul 164.

1. Moskovskiy gosudarstvennyy universitet.

(MIRA 17:10)

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

TECHNICATOR CONTROL TO THE SECTION OF THE CONTROL O

ZAYKOV, M.A.; TSELUYKOV, V.S.; KAMINSKIY, D.M.; KUZNETSOV, A.F.;
BELINSKIY, Ye.D.; SHAMETS, Ya.V.; FEDGROV, N.A.; BARITSKIY,
S.I.; ZAKHAROV, A.I.; ZHURAVLEV, M.A.; KOBYZEV, V.K.

Investigating energy and power parameters in plate rolling on reversing mills. Izv. vys. ucheb. mav.; chern. met. 7 no.2:100-107 164. (MIRA 17:3)

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

ZAKHAROV, A.I.

Effect of radiation on the physical properties and structure of a solid body. Dos. such. fiz. no.5:150-194 157.

(MIRA 16:6)

(Solids, Effect of radiation on)

Li

10.

30

20-114-6-16/54

AUTHORS:

Zakharov, A. I., Maksimova, O. P.

TITLE:

Martensite Transformations as Influenced by Bombardment With Neutrons (Vliyaniye neytronnogo oblucheniya na martensitnoye

prevrashcheniyè)

PERIODICAL:

Doklady Akademii Nauk SSSR,1957,Vol.114,Nr 6,pp.1195-1198(USSR)

ABSTRACT:

The present paper represents the first attempt to use the bombardment with neutrons in the study of martensite transformations. Steels and alloys were investigated in which the influence of the previous plastic deformation upon the martensite transformation was first thoroughly studied. The samples (2 x 3 x 24 mm) were bombarded in the active zone of a physical testing reactor with heavy water close to the uranium rods after previous (here described) heat treatment. One part of the samples was bombarded for 100 hours, the other part 200 hours with 1017 neutrons per cm2. The modification of the strength of the austenite was estimated from the form of magnetometric curves on deep cooling and on heating. Test resultsi Previous bombardment with neutrons exerts considerable influence upon the resistance of austenite to marten-

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特殊 国家组织的对方形式的主义和《马勒尔》:"等种的中国,但是目标的方面可用是和的现代使和理解的国籍的政治和自然也是由他的主动的政治和政治和政治的

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Martensite Transformations as Influenced by Bombardment With Neutrons

site transformation. The modification of resistance varies according to different materials. In steels the bombardment (independent of the carbon content and of the character of the alloys) increases the intensity of martensite transformation on subsequent deep cooling. In iron-nickel-manganese (N23G3 and N22G3) alloys with no content of carbon the bombardment always exerts a stabilizing influence on the γ -phase. The influence of bombardment and the influence of plastic deformation have much in common. If the bombarded samples are left lying for a long time at room temperature, the resistance of austenite is increased. On bombardment structural changes take place in the metals and alloys which influence the resistance of austenite in opposite directions. The total action (activating and retarding) depends on the total flux of neutrons and on the peculiarities of the material. There are 3 figures and 16 references, 9 of which are Slavic.

ASSOCIATION:

Central Scientific Research Institute for Errus Metalhingy (Tsentralnyy nauchno-issledovatel skiy institut cherncy metallurgii)

PRESENTED:

February 11, 1957, by G. V. Kurdyumov, Member of the Academy

Card 2/3

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

20-114-6-16/54 Martensite Transformations as Influenced by Bombardment With Neutrons

SUBMITTED: November 21, 1956

Card 3/3

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

ZAEHAROV, A.I., E-nd Flya-eth Sci -- (dies) "The effect of no tranradiction on the transformation of muctanite into martenaite." Inc. 1958. 11 pp (Fin of Higher Education USSR. For mainearing-Phys Inst), 120 coples (KL, 46-98, 137)

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"APPROVED FOR RELEASE: 03/15/2001

CIA-RDP86-00513R001963520005-0

SOV/137-58-8-17677

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr 8 p 211 (USSR)

Zakharov, A.I., Maksimova, O.P. AUTHORS:

Employment of Neutron Irradiation in Studying Martensite TITLE:

Transformations (Primeneniye neytronnogo oblucheniva dlya

issledovaniya martensitnogo prevrashcheniya)

PERIODICAL: Sb. tr. Inct metalloved, i fiz. metallov Tsentr. n. i. in-ta chernoy metallurgii, 1958, Vol 5, pp 124-135

The effect of neutron irradiation (N!) on martensite trans formation (T) was investigated on five types of steel containing ABSTRACT: respectively: 0.48% C, 7.7% Mn, 2.2% Gu (steel 50G8);

1.4% C, 4.0% Mn (steel 140G4); 0.50% C, 21.0% Ni (steel 50N21); 0.025% C, 22.7% Ni, 2.88% Mn (steel N23G3); 0.020% C, 22.4% Ni, and 3.48% Mn (steel N22G1). The effect of NI was evaluated by the change of progress of the martensite curves during cooling of specimens to a temperature of -196°C and heating to a temperature of 20°. The specimens were subjected to NI in the active zone, near the U rods, of an experimental physical heavy water reactor for

periods of 100 and 200 hours. Following the NI the specimens Card 1/2

CIA-RDP86-00513R001963520005-0" **APPROVED FOR RELEASE: 03/15/2001**

SOV/137-58-8-17677

Employment of Neutron Irradiation in Studying Martensite Transformations

were allowed to age at room temperature. The NI significantly influences the position of the martensite point and the over all T effect; the latter also depends on the type of material in question. In the case of carbon steels, the NI elevates the martensite point and increases the T effect during deep cooling. In the case of carbon-free alloys, the NI affects austenite in an opposite fash ion viz. the martensite point is lowered and the intensity of T during cooling In carbon steels, a certain amount of martensite is formed already during the NI process. The manner in which irradiation affects martensite T has much in common with the effect of plastic deformation. The aging of specimens at room temperature results in improved stability of austenite. The poor temperature stability of the activating effect of NI indicates that it is governed by formation of defects of the vacancy interstitial-intrusion type which produce elastic deformations in the crystal lattice. The stabilizing effect of NI is brought about by the formation of defects that are caused by division and disorientation of crystals, as a result of which the chances for the appearance and growth of martensite crystals are diminished.

1. Martensite--Transformations 2. Martensite-- M. Sh. Effects of radiation 3. Neutrons--Metallurgical effects

Card 2/2

ZAKHAROV, HIL

SOV/137-58-8-17644

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr 8, p 207 (USSR)

AUTHORS: Maksimova, O. R., Zakharov, A. I.

TITLE: On the Laws Governing the Elimination of Radiation Damage Upon

Annealing. (A Survey) [O zakonomernostyakh ustraneniya radia

tsionnykh narusheniy pri otzhige. (Obzor)]

PERIODICAL: Sb. tr. in-t metalloved, i fiz. metallov. Tsentr. n. 4, in ta

chernoy metallurgii, 1958, Vol 5, pp 528-549

ABSTRACT: A review of the laws governing the elimination of radiation

damage upon the annealing of metals. The character of and the laws governing the complex modifications of physical and mechanical properties and phase transformations in metals occurring upon irradiation, and the processes and the laws governing the restitution of the initial properties to metals upon annealing were examined together with an analysis of modifications occurring in five temperature ranges. Bibliography: 37 references.

1. Metals-Effects of radiation

2. Motals--Heat treatment

V. A.

Card 1/1

AUTHORS:

Zakharov, A.I. and Maksimova, O.P. (Moscow)

TITLE:

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation (Ob iznenenii kinetiki martensitnogo prevrashcheniya pod vliyaniyem oblucheniya)

PERIODICAL:

Izvestiya Akademii nauk SSSR, Otdeleniye tekhnicheskikh nauk, 1958, Nr 7, pp 3 - 9 (USSR)

ABSTRACT:

So far, in the theory of martensitic transformation the problem of the nature of loci of germination of a new phase has not been clarified. Also, the process of formation of martensite germinations involves such phenomena as incomplete martensitic transformation and austenite stabilisation. Investigations aimed at elucidation of the nature of the process of formation of martensite germinations are of particular importance from the point of view of the development of the theory of martensitic transformations. The development of nuclear techniques has provided new possibilities for creating various defects in the crystal lattice. Irradiation by means of fast particles may produce more elementary disturbances in the structure than can be obtained otherwise. In

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507/24-58-7-1/36

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

earlier work (Refs 1,2), the authors applied neutron irradiation in investigating martensitic transformations. They established that preliminary neutron irradiation, with a dose of the order of 10^{17} n/cm², has a considerable influence on the stability of the austenite and on the martensite transformation. This change in the stability differed with the material. In steels, irradiation brought about an intensification of the martensitie transformation during subsequent deep cooling with a constant speed. In carbon-free iron, nickel, and in Mnalloys, irradiation always had a stabilising effect on the Y-phase; it brought about & decrease of the martensitic point and of the transformation intensity. The aim of the work described in this paper was to investigate the influence of preliminary neutron irradiction on the kinetics of isothermal martensitic transformation at various temperatures. Furthermore, the authors aimed at elucidating the phenomena of eliminating the after effects of the radiation during annealing with a gradually increasing

Card 2/8

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

temperature. For solving the problem the authors used thermomagnetic, microstructural, X-ray and microhardness investigations. The experiments were carried out on 2.5 x 3.5 x 24.5 mm specimens of the alloy N22GZ (0.02% 0, 22.4% Ni, 3.48% Mn, Tm = -15 C) after homogenisation annealing at 1 150 C in vacuum for 10 hours, followed by removal of the surface layer to a depth of 0.25 mm. The irradiation was effected in the active zone of a reactor in the neighbourhood of the uranium rods. During irradiation the temperature of the specimens increased by no more than 40 C; the integral flux of the neutrons equalled 6.5 10 m/cm². As can be seen from Figure 1, the stabilisation effect of this dosage of irradiation was about twice as intensive as in earlier experiments in which a flux intensity of 4.2 10 m/cm² was used. In Figure 1 the martensitic transformation curves are graphed for a specimen which has been irradiated and also for one which has not been irradiated. In Figure 2 the curves of iscthermal martensitic transformation at various temperatures are graphed for the irradiated and non-irradiated states.

Card 3/8

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On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

In Figure 3, the curves of isothermal martenaitic transformation for irradiated and non-irradiated austenite are graphed for the temperatures of -25 and -50 °C. In Figure 4, the initial speed of isothermal martensitic transformation, as a function of the temperature, is graphed for irradiated and non-irradiated austenite. graph, Figure 6, shows the influence of annealing on the stability of preliminarily irradiated austenite. In Figure 7, the changes are graphed of the martensitic point and of the microhardness during annealing of irradiated austenite. In Figure 5, misrostructure photographs are reproduced of the martensite which formed in irradiated and non-irradiated specimens after isothermal holding for 7 hours at -50 °C. The results are in agreement with those obtained during earlier investigations relating to the influence of irradiation on the kinetics of martensitic transformation (Ref 1). The relations determined earlier (Ref 2) were confirmed and new relations were established which give a better understanding of the disturbances to

Card 4/8

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

which the changes in the kinetics of martensitic transformation are attributed. The authors conclude that as a result of irradiation, defects of two distinct types are generated in the γ -phase which influence differently the stability of the austemite relative to that of the martensite. Iow-stability defects bring about activation of the austenite. On the other hand, stabilisation of the γ -phase is due to the occurrence of radiation disturbances which possess a high stability. Elimination of the radiation effects of activation during storage and during annealing at relatively low temperatures (20-200 °C) in steels and further intensification of the stabilisation observed under the same conditions in alloys are phenomena which can be attributed to the removal of radiational disturbances of the same type. Obviously, these disturbances are "defects" which bring about an increase in the electric resistance of the irradiated metals. Such a conclusion can be derived from the results described in this paper am from analysis of literary data which indicate a coincidence of the temperature ranges of radiation effects during

Card 5/8

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

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On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

annealing. The increase of the electric registance during irradiation is due mainly to the occurrence of elementary defects of low stability of the penetration-atom type, which can be easily eliminated by recombination during which can be easily eliminated by recombination during storage and during low-temperature annealing. Thus, as a result of annealing, a decrease should take place in the concentration of the pair defects in the volumes affected concentration of the pair defects in the assumption that by "thermal peaks". On the tasis of the assumption that elastic distortions, brought about by "thermal peaks" and possessing a sufficiently high concentration of pair possessing a sufficiently high concentration of pair defects can lead to the germination of martensified the phenomenon of reduced activation and intensified stabilisation of the austenite during low-temperature annealing is understaniable. It was found that elimination of the increased stability of the austenite in the irradiated alloy N22GZ will begin during annealing in the temperature range above 200 C and this process is terminated on heating to 600-700 C. It is known that the inated or heating to 600-700 C. It is known that the

Card 6/8

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

temperature range of removal of the effect of hardening brough about by irradiation (Ref 7). In the given case, the main part of the hardness increase of the irradiated the main part of the hardness increase of the irradiated temperature range 200-500 C. Consequently, the process of temperature range 200-500 C. Consequently, the process of temperature range 200-500 C. Consequently, the process of temperature range and of the re-establishment of the mechanical properties and of the stabilisation effect of the radiation of the austenite for the austenite in the same temperature range. In proceeds during heating in the same temperature range. In the fine crystalline structure of the austenite, which the fine crystalline structure of the austenite, which the fine crystalline structure of the austenite, which alarge extent responsible for the observed stabilisation of the irradiated austenite. Accordingly, the stabilisation effect of the radiation can be explained by the limitation of the growth of the martensite crystals in the distorted and hardened matrix.

card 7/8

SOV/24-58-7-1/36

On the Changes in the Kinetics of Martensitic Transformation as a Result of Irradiation

There are 7 figures and 8 references, 6 of which are Soviet and 2 English.

SUBMITTED: April 19, 1958

Card 8/8

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

5/137/62/000/004/138/201 A060/A101

AUTHOR:

Zakharov, A. I.

TITLE:

Determination of the total neutron flux under irradiation of

materials in a nuclear reactor

PERIODICAL: Referativnyy zhurnal, Metallurgiya, no. 4, 1962, 92, abstract 41555

("Sb. tr. In-t metalloved. i fiz. metallov Tsentr. n.-i. in-ta

chernoy metallurgii", 1959, 6, 389-393)

A method is described for determining the absolute intensity of neutron flux using scintillators. A calculation of activation is given, as well as recommendations for the choice of scintillators. An analysis is given of methods of identifying the radiations from their hardness by the use of a Cd filter. A concrete example is given for the use of a W-scintillator. The methods described make it possible to take account of intrinsic absorption in the scintillator, absorption in the air, and the window material of the counter, reflection from the backing, etc. The absolute precision of the determination of total intensity is estimated to be of the magnitude of ~ 50%. It is noted

Card 1/2

"APPROVED FOR RELEASE: 03/15/2001

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S/137/62/000/004/138/201 A060/A101

Determination of the total neutron ...

that for comparative measurements it is possible to attain a precision of up to 1 - 26.

N. Coveling

[Abstractor's note: Complete translation]

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o) interior	runrozz: This book is intended for astallurgists, meaning and exectalises in the physics of metals. Coverand: The papers in this collection present the results of coverand: The papers in this collection present the results of coverand: The papers in this collection present in 1996. Subjects investigations conducted between 1994 and 1996. Subjects for a conduction of setals, physical methods in incluencing the processes of crystallisation, problems in the incluencing the processes of crystallisation, problems in the production control. References follow each efficie. TABLE OF CONTENTS:	Problems in Physical Motalinian installations for levitation Filake, Tu.5. Method of Designing Installations for levitation belief of Settle. Religie action of levitation selling of settle are compared, and a method of designing an inductor sufficiently accurate leving seaton of levitations is proposed. Fermina, I.T. Frinciples of Designing Ragnetostristive withration bears principles of designing Ragnitostrictive withration Physikators of designing Ragnitostrictive withration bears principles of designing Ragnitostrictive withration bears greening in induced inductive and induced as presented. Special fee univaries of special as given to the analysis of operating conditions as given to the analysis and alloys in section for a linguistic for Type for a Microphotoseter greening of the analysis for a Microphotoseter franchism of the analysis for a Microphotoseter franchism and E.S. Tokmakov. ***Association of Method Used in Metalizer Doron metalizer in the section for the section fethod Used in Metalizer in the section for the section fethod Used in Metalizer in the section for the section fethod Used in Metalizer in the section fethod in the section fethod in the section fethod in the section fethod	The state of the control of the control of the control of the control of a colt of the colt of th	ys. 5/16
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Card 1/2

5/070/62/007/003/020/026 E132/E460

Ponyatovskiy, Ye.G., Zakharov, A.I.

On the question of the crystal structure of the high

AUTHORS:

temperature modification of thallium TITLE:

PERIODICAL: Kristallografiya, v.7, no.3, 1962, 461-463 The results of this paper were presented at the 7th Scientific-Technical Conference for the use of X-rays for A polycrystalline film of Tl, the surface of which had been mechanically freed from oxide, was examined in an X-ray diffractoinvestigation of materials. meter with Cu radiation at temperatures between -190°C and the melting point of Tl. On first heating up to 232°C; the h.c.p. structure was found up to this temperature where the alpha to beta transformation took place very sharply in less than a second Recrystallization rapidly took place, big The structure was then b.c.c. Further cycles through the transformation did not reduce the grain size. To avoid these grain size effects a special specimen of fine grains (heating 1.5°/min). mixed with aluminium filings was prepared.

On the question of the crystal ...

2. 在中国的基础的研究。

s/070/62/007/003/020/026 E132/E460

 β -T1 had a = 3.871 \pm 0.002 Å. Discrepancies in the literature as to whether the high-temperature form was f.c.c. or b.c.c. are satisfactorily resolved. There are 3 figures.

ASSOCIATION: Tsentral'nyy nauchno-issledovatel'skiy institut

chernoy metallurgii im. I.P.Bardina

(Central Scientific Research Institute for Ferrous

Metallurgy imeni I.P.Bardin)

SUBMITTED:

October 26, 1961

Card 2/2

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

S/078/62/007/010/003/008 B144/B186

AUTHORS:

Zakharov, A. I., Ponyatovskiy, Ye. G.

TITLE

Phase diagram of thallium - tin alloys

PERIODICAL: Zhurnal neorganicheskoy khimii, v. 7, no. 10, 1962, 2374-2377.

TEXT: A supplementary phase diagram of T1-Sn allyos containing up to 15 at.-% 3n (Fig. 3) was plotted for the temperature range from 20°C up to the melting point in order to elucidate the inconsistencies between, on the one hand, the previous data of the present authors (Kristallografiya, 461 (1962)) and of H. Lipson, A. R. Stokes (Rature, 146, A37. (1941)), and on the other hand, the data of M. Hansen (Constitution of Binary Alloys, N. Y., 1958, p. 1214) and of J. C. Blade, E. C. Ellwood (J. Inst. Met., 88, 186 (1959)). X-ray diffraction patterns of 10 different alloys were taken at different initial phase change temperatures. The temperature dependence of the intensity of line (102) of the hexagonal phase, and that dependence of the face-centered cubic phase, were recorded in addition of line (002) of the face-centered cubic phase, were recorded in addition to complete X-ray pictures. The patterns of an alloy containing 4.91% Sn to complete X-ray pictures. The patterns of the sample results in sutectic

Card 1/4 2

5/078/62/007/010/003/008 B144/B186

Phase diagram of thallium - tin alloys

decomposition of the $\alpha+\delta$ phase and in formation of the β phase. There are 4 figures and 1 table.

A3500IATION: Tountral'nyy nauchno-issledovatel'skiy Institut chernoy

metallurgii (Central Scientific Research Institute of Ferrous

Metallurgy)

SUBMITTED:

January 4, 1962

Fig. 3. Phase diagram of TI-Sn alloys rich in TI.
Legend: (1) hexagonal dense packing, a phase; (2) face-centered cubic lattice, δ phase; (3) body-centered cubic lattice, β phase; (4) c+δ; (5) interface of the appearance of the β phase; (6) interface of the appearance of the liquid phase; (a) at.-%; (b) % by weight; full lines with experimental points; interfaces based on the authors' results; full lines without points; data of Blade and Ellwood; broken lines; suggested interfaces.

Card 2/6 2

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ACCESSION NR: AP4042559

s/0056/64/046/006/2003/2010

AUTHOR: Zakharov, A. I.; Kadomtseva, A. H.; Levitin, R. Z.; Ponyatovskiy, Ye. G.

TITLE: Hagnetic and magnetoelastic properties of a metamagnetic iron-rhodium alloy

SOURCE: Zh. eksper. i teor. fiz., v. 46, no. 6, 1964, 2003-2010

TOPIC TAGS: magnetostriction, alloy Young modulus, alloy lattice parameter, ferromagnetic transition temperature, Curie point, iron rhodium alloy, alloy magnetization, Alloy

ABSTRACT: The temperature dependences of the magnetization, magnetostriction, Young modulus, and lattice constant of an iron-rhodium alloy of close to equiatomic (Fe_{0.5}, Rh_{0.5}) composition have been investigated in the 50-750K temperature range. The experiments were conducted on vacuum-melted Fe-Rh alloy annealed at 1100C for 5 hr and then furnace cooled or water quenched from 1100C. In a field up to 2000 oe, the annealed alloy was antiferromagnetic at room temperature, with the transition to the ferromagnetic state occurring in a

Card 1/3 ---

ACCESSION NR: AP4042559

field of 1700 oe at 358K with heating and at 352K with cooling. The Curie point of the alloy, determined in a 9-oe field, was about 660K. The transition temperature Tk was found to decrease by about 12K, with the field increasing to 14,500 oe. Isothermal curves for the magnetization in fields up to 140 koe showed that below the critical temperature T_k , the magnetization increases sharply in certain critical fields Hk, i.e., the antiferromagnetic-to-ferromagnetic transition occurs under the action of the field. The critical field Hk, defined as the field magnitude at which the most rapid increase in magnetization occurs, decreases linearly with increasing temperature at a rate of 0.0017 oa/deg. The lattice parameter increases gradually with the temperature increase to T_{k} " 353K, at which a new ferromagnetic phase is formed whose lattice parameter increases abruptly by 0.3%. Above the Curie point (0 = 660K), the lattice parameter increases with temperature more rapidly than in the ferromagnetic region. With an increasing hydrostatic pressure, the transition temperatures, both in heating and cooling, increase approximately linearly at a rate of 0.00433 deg/atm. The Young modulus exhibits a sharp increase at the point of transition from the antiferromagnetic to the ferromagnetic state. The longitudinal magnetostriction & and the relative change

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ACCESSION NR: AP4042559

of Young modulus AE/E are zero in the antiferromagnetic region but are at a maximum in the region of temperature transition. The maximum probably results from the superimposition of magnetoelastic effects, which are associated with the destruction of the antiferromagnetic structure under the action of the field, on the ordinary AE and magnetostriction effects which are caused by domain processes. The use of the data obtained for determining the applicability of the C. Kittel theory to ferromagnetism — antiferromagnetism transition in the Fe—Rh alloy produced inconclusive results — and further research on the alloy is recommended. Orig. art. has: 8 figures.

ASSOCIATION: Moskovskiy gosudarstvennyy universitet (Moscow State University)

SUBMITTED: 18Jan64

ATD PRESS: 3068

ENCL: 00

SUB CODE: MM.SS

NO REF SOV: 006

OTHER: 009

Card 3/3

	Phase diagram of thallium-tin alloys. Zhur.neorg.khim. 7 no.10:2374-2377 0 62. (MIRA 15:10)
	1. TSentral'nyy nauchno-issledovatel'skiy institut chernoy metallurgii. (Thallium-tin alloys)
* .	

FONYATOVSKIY, Ye. G.; ZAKHAROV, A. I.

Grystalline structure of a high-temperature modification of thallium. Kristallografiia 7 no.3:461-463 Ky-Je '62. (MIRA 16:1)

1. TSentralinyy nauchmo-issledovateliskiy institut chernoy metallurgii imeni Bardina.

(Thallium) (X-ray crystallography)

39022 Z/009/62/000/001/001/001 E073/E335

11.1260

AUTHORS:

Macharacek, K., Zakharov, A.I. and Aleshina, L.A.

TITLE:

Heats of combustion and formation of isomeric

dinitroanilines

PERIODICAL: Chemický průmysl, no. 1, 1962, 23 - 24

TEXT: The heats of combustion of all isomeric dinitroanilines were measured at constant volume and from the obtained values the heats of combustion at constant pressure and the heats of formation at constant volume and pressure were calculated. The values (averages of three measurements) obtained for the molar heats of combustion and formation (kcal/mole) are given in Table 2. There are 2 tables.

ASSOCIATIONS:

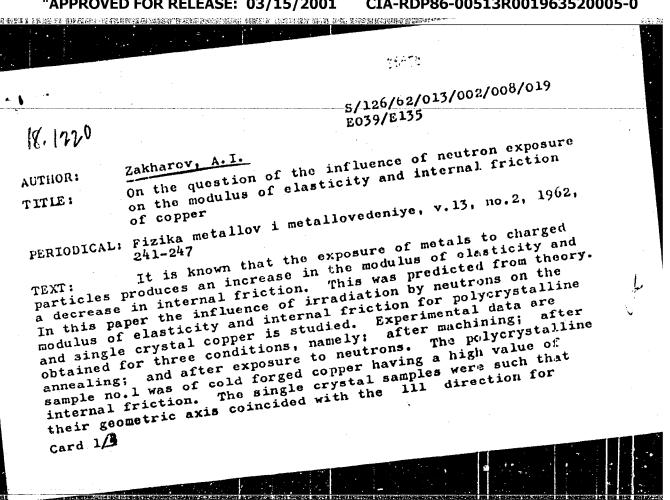
Ústav teoretických základu chemické techniky ČSAV, Praha (Institute of Theoretical Fundamentals of Chemical Engineering, ČSAV,

Prague)

Lensovet Leningrad Technological Institute,

Leningrad, USSR.

Card 1/2



CIA-RDP86-00513R001963520005-0" APPROVED FOR RELEASE: 03/15/2001

On the question of the influence ... E039/E135

sample no.2 and with the 100 direction for sample no.3. A resonance method was used for obtaining the experimental data and the results are given in Table 2, showing how the resonant frequency and modulus of elasticity change with the form of treatment. Samples were annealed at 600 °C for 2 hours under vacuum and irradiated for 10 minutes in a thermal neutron flux of 1.5 x 1013 n/sec containing 10% neutrons of 103 eV energy. The dependence of the internal friction on the amplitude of bending oscillations is shown graphically. Sample no. 2 in the annealed condition had a much larger value of internal friction than sample no.3. After exposure to neutrons the internal friction was reduced to almost a half its previous value for sample no.2 but there was no change in sample no.3. The conclusions drawn are as follows. Exposure of polycrystalline and single crystal copper to neutrons increases the modulus of elasticity and reduces plastic deformation and internal friction. In addition it causes a displacement of the threshold amplitude dependence on internal friction in the direction of large amplitude oscillations. The effect of exposure is removed by Card 2/4

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

On the question of the influence... 5/126/62/013/002/008/019 E039/E135

annealing in the range of the temperature of recrystallisation. In the polycrystalline sample the influence of radiation is similar in character to the effect of a small admixture. The nature of the dislocations produced by radiation is discussed in detail.

There are 6 figures and 2 tables.

SUBMITTED: Initially, July 26, 1960.
After revision, June 17, 1961.

Card 3/4

29605

55800

S/120/61/000/004/015/034 E202/E592

AUTHOR:

Zakharov, A.I.

TITLE:

X-ray diffractometer

PERIODICAL: Pribory i tekhnika eksperimenta, no. 4, 1961, 109-113

The author describes an X-ray diffractometer designed primarily for work with polycrystalline samples in the shape of thin plates, but capable also of work with other shapes and single crystals. Very simple adjustment to specimen holder permits this instrument to be used over a very wide temperature range viz. from -196 to +600°C. The geometry of the system is shown in Fig.1. The diffracting surface of the flat sample 1 is coplanar with the horizontal plane, while the X-ray tube 2 and the detecting crystal 4 traverse synchronously in the opposite directions around the axis contained in the plane of the sample. Prior to its entry to the detector, the diffracted beam falls on the bent quartz monochromator 3, which changes the direction of the beam and thus allows this system to be used in the study of radioactive samples. A series of vacuum locks permit changing the sample without stopping the oil diffusion pump. The sample itself is in Card 1/4

29505

X-ray diffractometer

Card 2/4

S/120/61/C00/004/015/034 E202/E592

tight thermal contact with the thermostat or cryostat and, for high temperature work, these may be replaced by a heater. For Debye-Shearer work the sample may also be rotated. The rotation of the X-ray tube and the monochromator with the scintillation counter arms may be either manual or by means of a synchronous motor geared for three fixed velocities viz. 2; 1 and 0.5 deg/min. The angles of turn of the tube and detector are read from a vernier to 1'. The HV is applied to the cathode of a 308-3 (VSV-3) or VSV-6 X-ray tube from a HV transformer by means of a flexible cable. The recording installation has two channels. The main channel registers the diffraction spectrum using a scintillation counter of the type described by B. N. Vasichev, V.A. Il'ina, V.K.Latyshev and Yu.S. Pliskin (Ref. 1: PTE, 1960, No.2, 51). The scintillations are registered by two photomultipliers which are followed by a coincidence circuit and a single channel differential amplitude analyser which rejects all the impulses with amplitudes which are not derived from the characteristic radiation of the target (e.g. cosmic rays, radioactivity, secondaries). The intensity of the selected impulses is evaluated and either read directly or

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X-ray diffractometer

5/120/61/000/004/015/034 E202/E592

registered continuously and recorded graphically. The second (subsidiary) channel registers the stability of the primary beam. The latter is made to pass through a thin Al foil 8, which scatters some of the beam and some of this scattered radiation enters a G.M. counter. Any variations in the intensity of the primary beam are accompanied by a proportional variation in the intensity of the scattered beam reaching the counter, which in turn may be regulated and registered, e.g. by feeding the suitably amplified output to the stabiliser of the anode current of the X-ray tube. In the particular design quoted, the anode current stabilisation is achieved by varying the heater voltage on the cathode of the X-ray tube. Details of such a circuit are given in the paper of M. A. Blokhin, I. V. Busler, O.P. Kramarov and I. P. Chernyavskaya (Ref. 2: PTE, 1959, No.1, 106). The author does not give detailed performance figures of the diffractometer. but two contrasting examples are quoted: one showing the heights of the intensity peaks of (0002) line of thallium at -196 and +20°C. and the other illustrating the intensity curve due to the same line over a range covering the transformation from the (α)Tl phase \Rightarrow There are 5 figures and 2 Soviet references. (B)T1. Card 3/4

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

1950% X-ray diffractometer 5/120/61/000/004/015/034 E202/E592 ASSOCIATION: Tsentral'nyy nauchno-issledovatel'skiy institut chernoy metallurgii (Central Scientific Research Institute of Ferrous Metallurgy) SUBMITTED: October 21, 1960 Fig.1. Legend. Geometry of the diffractometer. 1 - sample: 2 - X-ray tube; 3 - monochromator; 4 - scintillator (detector)NaI(T1) crystal; - photomultiplier: 6 - G.M. tube; 7.9 - collimators; 8 - A1 foil; 10 - camera: 11 - thermostat. Card 4/4

PHASE I BOOK EXPLOITATION

SOV/6176

Konobeyevskiy, S. T., Corresponding Nember, Academy of Sciences
USSR, Resp. 2d.

Deystvive vadernykh izlucheniv na materialy (The Effect of
Nuclear Radiation on Materials). Moscow, Izd-vo An SSSR,
Nuclear Radiation on Materials and SSSR.
Nuclear Radiation on Materials and SSSR.
Otdeleniye tekhniSponsoring Agency: Akademiya nauk SSSR. Otdeleniye tekhnicheskikh nauk; Otdeleniye fiziko-matematicheskikh nauk.
Otheskikh nauk; Otdeleniye fiziko-matematicheskikh nauk.

Resp. Ed.: S. T. Konobeyevskiy; Deputy Resp. Ed.: S. A.
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Housei H. G. Makarenko; Tech. Eds: T. V. Folyakova and
I. N. Dorokhina.
Card 1/14

The Effect of Muclear Radiation (Cont.)

PURPOSE: This book is intended for personnel concerned with nuclear materials.

COVERAGE: This is a collection of papers presented at the Moscow Conference on the Effect of Nuclear Radiation on Materials, held December 6-10, 1960. The material reflects certain trends in the work being conducted in the Soviet certain trends in the work being conducted in the Soviet scientific research orginization. Some of the papers are devoted to the experimental study of the effect of neutron devoted to the experimental study of the effect of neutron irradiation on reactor materials (steel, ferrous alloys, irradiation on varial, graphite, and nichromes). Others deal molybdenum, avial, graphite, and nichromes of internal streames, chemical transformations, relaxation of internal streames, chemical transformations, relaxation of internal streames, internal friction) and changes in the structure and properties of various crystals. Special attention is given to ties of various crystals. Special attention is given to ties of various crystals. Special attention is given to the effect of intense y -realistion on the electrical, magnetic, and optical properties of metals, dielectrics, and semiconductors.

Card 2/14

APPROVED FOR RELEASE: 03/15/2001 CIA-RDP86-00513R001963520005-0"

The Effects of Nuclear Radiation (Cont.)	30V/ 6176
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Andronikashvill, R. L., N. G. Politov, and L. F. Vorozheyki Effect of Lattice Disturbances on Medianical and Optical Properties of Potassium Chloride Crystals	
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S/142/60/000/003/002/017 E192/E482

AUTHOR:

Zakharov A.L.

TITLE:

Equivalent Circuit of a Spacistor Amplifier

PERIODICAL: Izvestiya vysshikh uchebnykh zavedeniy, Radiotekhnika,

1960, No.3, pp.309-318

TEXT: The analysis given in this article represents a continuation of the author's earlier work (Ref.2). The idealized model developed by the author is employed and it is assumed that the expressions for the parameters µ (low frequency voltage amplification coefficient) and R_i (low frequency internal impedance which does not take into account the resistance of the electrode material) are known (Ref.2). Various feedback currents and leakages are neglected and only the injection of the carriers is taken into account. It is assumed that the system operates linearly, that is very small sinusoidal signals are The velocity of the carriers is assumed to be constant, so that the signal of the conduction current is produced by the movement of a small quantity of additionally injected carriers. The emitter current component due to the displacement current is Card 1/5

86789 \$/142/60/000/003/002/017 E192/E482

Equivalent Circuit of a Spacistor Amplifier

expressed by

$$I_{cm, 5} = -\alpha I = -\frac{i\omega \Phi_{3} e}{4}$$
 (1)

where $\phi_{\mathfrak{D}}$ is the flux of the electric field vector entering the emitter and α is a coefficient determining the portion of the emitter current which is produced by the displacement current. The collector current is determined by the injection current during time interval from $t-\tau$ to t, where τ is the transit time of the carriers through the depletion layer. This current is expressed by

$$I_K(t) = -\frac{1}{\tau} \int_0^1 I(t-9)d\theta$$
 (2)

The collector current can also be represented as a difference between the injection current I and a quantity βI . The Card 2/5

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